Resonance-assisted light–control–light characteristics of SnS$_2$ on a microfiber knot resonator with fast response

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An all-optical light–control–light functionality with the structure of a microfiber knot resonator (MKR) coated with tin disulfide (SnS$_2$) nanosheets is experimentally demonstrated. The evanescent light in the MKR [with a resonance $Q$ of $\sim 59,000$ and an extinction ratio (ER) of $\sim 26$ dB] is exploited to enhance light–matter interaction by coating a two-dimensional material SnS$_2$ nanosheet onto it. Thanks to the enhanced light–matter interaction and the strong absorption property of SnS$_2$, the transmitted optical power can be tuned quasi-linearly with an external violet pump light power, where a transmitted optical power variation rate $\Delta T$ with respect to the violet light power of $\sim 0.22$ dB/mW is obtained. In addition, the MKR structure possessing multiple resonances enables a direct experimental demonstration of the relationship between resonance properties (such as $Q$ and ER), and the obtained $\Delta T$ variation rate with respect to the violet light power. It verifies experimentally that a higher resonance $Q$ and a larger ER can lead to a higher $\Delta T$ variation rate. In terms of the operating speed, this device runs as fast as $\sim 3.2$ ms. This kind of all-optical light–control–light functional structure may find applications in future all-optical circuitry, handheld fiber sensors, etc.

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1. INTRODUCTION

Modern high-performance systems are built by the combination of electronic and photonic components in order to take advantage of each [1,2]. Among different photonic components, a resonant-based structure such as a microfiber knot resonator (MKR) is an important one that finds applications in sensors [3], lasers [4], filters [5], etc. An MKR is a microscopic loop elastically bent by submicrometer silica wire usually with a loop diameter of several or hundreds of micrometers. The sub-micrometer silica wires can be easily obtained from a tapered microfiber (MF), which has a thinned core size that enables light leakage outside the waveguide core [6,7]. Particularly, the large fraction of evanescent light, when it is combined with materials whose property can be tuned by light [8–10], can achieve a large panel of all-optical devices with different functionalities [11].

Two-dimensional (2D) materials are excellent candidates to be combined with fiber-optic components since their properties can be tuned by light and their atomically thin thickness can facilitate structure fabrications. In addition, there are whole large 2D materials families such as monoelemental graphene [12,13], antimonene [14–16], and phosphorene [17,18]. Few-layer antimonene-decorated MF employed as an optical saturable absorber for ultrafast photonics operation and a stable all-optical pulse thresholder is demonstrated in Ref. [14]. Recent reports show that it can also be employed as an all-optical Kerr switcher and wavelength converter where modulated high-speed signals at a frequency up to 18 GHz are achieved [15], and it presents high stability under ambient conditions that can last for months [16]. New 2D material such as phosphorene is reported in the application of a robust delivery platform for cancer theranostics and development of reliable devices for optoelectronic applications [17,18]. There are also compounds with the form of MX$_2$, where M stands for the transition metal and X stands for the dichalcogenide element [19] and 2D layered metal dichalcogenide (LMD) such as
SnS$_2$, MoS$_2$, and WS$_2$. However up until now, there is no single material that is developed into the sole dominant material for the optoelectronic and photonic applications. This is due to the fact that various materials have different physical and chemical properties in terms of stability, electron mobility, on/off ratio, thickness-dependent bandgaps, etc.

Tin disulfide (SnS$_2$), which belongs to the extended families of LMD, has unique properties such as large surface areas, high on/off ratio, finite bandgap of $\sim 2.35$ eV, strong absorption property in the visible regime, high discharged capacity, and high carrier mobility, making it a suitable material for developing next-generation electronics or photonics devices [21,22]. It has been investigated as field effect transistors [23], photodetectors [20,24,25], photocatalysts [26,27], solar cells [28], etc. Particularly, the property of its strong absorption can be exploited as light–controlled–light all-optical devices.

In this paper, a light–controlled–light functionality by the structure of an MKR with SnS$_2$ is demonstrated where the transmitted signal power is controlled via the violet pump light power. The largest transmitted power variation rate $\Delta T$ versus violet light power is about 0.22 dB/mW, while the structure can run as fast as 3.2 ms. In addition, the light amplitude tuning experiment in the MKR with a SnS$_2$ structure enables a direct demonstration that resonances with a larger $Q$ and a higher extinction ratio (ER) can yield a higher sensitivity. The paper is structured as follows: the structure fabrication of SnS$_2$–coated MKR is first elaborated, and then the experimental details, phenomenon, and discussion on the obtained results are presented. Simulations by the coupled mode theory are also carried out in order to uncover the physical mechanism of the observed phenomena. Afterwards, the response time of the device is measured. Lastly, the main results are compared with other types of structures, and some perspectives are provided.

2. DEVICE FABRICATION

In order to obtain the MKR with the SnS$_2$ structure, an MKR is first needed to be fabricated. The MF (which is utilized to form the MKR) fabrication starts with a standard SMF-28 (Corning) with a core diameter of 8 $\mu$m, and it is fabricated by the heat flame taper-drawing method [7]. Afterwards, the MF is assembled into an intertwined MKR structure with the aid of translational stages and microscopes. The fabricated MKR is then packed onto a MgF$_2$ crystal substrate with a high degree of cleanliness.

Microscopic imaging and measurement of the optical transmission are performed for the MKR structure characterization. The microscopic image of the structure is shown in Fig. 1(a), which depicts the MKR structure with a diameter of $D \approx 480.6$ $\mu$m. The inset in Fig. 1(a) shows the waist region of the MF with a diameter of $d \approx 7.0$ $\mu$m. We can see from these images that the MF has a low surface roughness and the MKR is assembled with good quality. Transmission is measured by connecting a tunable laser source (TLS) to one end of the MKR, while the other end is connected to an optical spectrum analyzer (OSA). Figure 1(b) shows the measured transmission spectrum of the MKR from which we can deduce that the MKR has a free spectral range (FSR) of $\sim 1.07$, a $Q$ factor of $\sim 40,586$, and an ER of $\sim 18.0$ dB at a resonance wavelength around 1542.3 nm.

The next step is to deposit the SnS$_2$ nanosheets onto the MKR, which has low-loss, high-$Q$ factor characteristics. The SnS$_2$ dispersions employed in our experiment are fabricated by the lithium ion intercalation exfoliation method with a concentration of 1 mg/mL. The SnS$_2$ nanosheets have finite lateral sizes of about 0.05–1 $\mu$m, while the thickness varies from 1 to 10 layers [29]. Raman and UV-Vis absorption spectra are performed for the characterization of SnS$_2$ nanosheets, and the results are shown in Fig. 2.

The Raman spectrum of SnS$_2$ shown in Fig. 2(a) depicts a peak around 313.8 cm$^{-1}$. It corresponds to the A$_{1g}$ mode of the SnS$_2$, which is the signature of SnS$_2$ interlayer molecular oscillation [23]. Notice that the usual detected weak intralayer E$_{g}$ mode in SnS$_2$ crystal Raman spectra is not found here in Fig. 2(a). It might be due to a too weak rejection of the Rayleigh-scattered radiation to be detected by the Raman sensor. The UV-Vis absorption spectrum is shown in Fig. 2(b), from which we can see that it has strong absorption at the wavelength ranging from 200 nm to 500 nm and a local peak around 250 nm. The absorption property is relatively strong at a wavelength of 405 nm, which corresponds to violet light [27].

After characterizing the SnS$_2$ dispersions, we then perform an ultrasonic treatment for $\sim 30$ min at a temperature of 25°C in order to obtain quasi-evenly distributed SnS$_2$ nanosheets. Immediately after the ultrasonic process is finished, a pipette is employed to transfer the dispersions to the arc areas of the MF away from the knot of the MKR. The reason for coating the SnS$_2$ away from the intertwined knot is that the resonance condition might not be satisfied since too much of the
absorption will lead to a small fraction of the light being recirculated back to the loop. In addition, coating only the areas away from the knot, one makes sure that the deposition of the SnS$_2$ nanosheets will induce an increased loss factor to the resonator [30]. Consequently, it mainly affects the transmission amplitude of the resonance.

The MKR with the SnS$_2$ structure is built after the solvent is evaporated, and it reaches a stable state (usually it takes about several hours). A TLS and an OSA are employed for transmission measurement, which serves as a sign of whether the evaporation is finished or not since the output spectrum will have little variation when it is stable. Figure 3(a) shows a microscopic image of the final fabricated device where about one third of the area away from the knot is coated with SnS$_2$. Figure 3(b) shows an scanning electron microscope (SEM) image of a small part of the MKR circumference from which we can see that the SnS$_2$ nanosheets are successfully coated onto the MKR structure where the thickness varies from 100 nm to 300 nm. Optical characterization and external vertical violet pump light for MKR resonance amplitude tuning will be presented in the following section.

3. EXPERIMENTAL DETAILS, RESULTS, AND DISCUSSION

In this section, optical characterization of the MKR with and without SnS$_2$ will be presented. The experimental setup for the device characterization is shown in Fig. 4, where the sample is fixed at a basin made of UV adhesive. Light from TLS (ANDO-AQ4321D) is connected to one end of the MKR structure while the output is connected to an OSA (YOKOGAWA-AQ6317C). The 405 nm laser diode (LASERV-LSR405NL) is placed vertically above the sample (either MKR with or without SnS$_2$) at a distance of $\sim$10 cm, and it is focused by a cylindrical lens. The focused light then shines directly to the MKR areas centering in the arc area away from the knot.

Figure 5(a) shows the transmission spectrum of the MKR structure with and without SnS$_2$ at the off-state of the violet pump light. Qualitatively, there are several noticeable distinctions between the two curves. First, the overall transmission of the MKR with SnS$_2$ [green curve in Fig. 5(a)] is $\sim$5.0 dB lower than that of the MKR without SnS$_2$ [red curve in Fig. 5(a)]. Second, the spectrum seems smoother in the case of the MKR with SnS$_2$ than in the case of MKR without SnS$_2$. Third, the resonance line shape is broadened in the case of the MKR with SnS$_2$ when compared with that in the case of MKR without SnS$_2$. As to the 5.0 dB in the transmission difference and the broadened resonance, these are mainly due to the increased loss factor brought by the deposition of SnS$_2$. In terms of the smoother curve, it indicates that there is only one dominant resonance condition satisfied in the MKR with SnS$_2$ structure while other possible resonances are suppressed due to the increased loss resulting from SnS$_2$. Similar phenomena of a smoother transmission curve after adding materials with absorption are also reported in Ref. [31]. In the case of the MKR without SnS$_2$, several resonances occurred. One dominant resonance, for which the mode-coupling efficiency is the highest among others, forms the comb-like shape of the resonance spectrum in the case of the MKR without SnS$_2$. Other minor resonances are shown as small dips in the spectrum as the one highlighted as the purple ellipse in Fig. 5(a). These are probably created by the slightly inhomogeneous properties of the MF with SnS$_2$ nanosheets. The inhomogeneous properties here refer to the quasi-periodically small diameter difference along the circumference of the MKR, which is caused by the manually pulling force pattern employed during the MF fabrication. Notice also that the $\lambda_{\text{res}}$ position does not overlap between the two curves in Fig. 5(a). If one compares the $\lambda_{\text{res}}$ corresponding to the maximum ER in the two cases, a red shift phenomenon in the MKR with SnS$_2$ will be observed. This indicates that the
resonance order and the mode effective index of the resonance in the two cases might not be the same.

Theoretically, there are ways of estimating the absorption induced by SnS$_2$, such as what is presented in Ref. [13], which is obtained via fitting the notch region of the spectrum. Simulations according to the coupled mode theory are performed to fit the experimental curves in Fig. 5(a). The comparison of the measured normalized transmission spectra with the fitted curves is shown in Figs. 5(b) and 6(a). The normalized experimental transmission spectra are normalized to the transmission obtained within the bare MF. The fitted curves are obtained according to the following equation based on the coupled mode theory [32,33]:

$$|T|^2 = \frac{(1 - \gamma) - 2\kappa_r[1 + \sin(\beta L)]}{1 + \kappa_r^2 + 2\kappa_r\sin(\beta L)},$$

where $\gamma$ is the coupling loss due to the light scattering by the twisted knot and the attenuation brought by the MKR loop, and $\kappa_r$ is the coupling coefficient that relates to the fractional coupling intensity at different ports of the MKR. The optimum value of the coupling depends mostly on the parameters of loss and the coupling coefficient. $\beta = (2\pi/\lambda)\text{Re}(n_{\text{eff}})$ is the propagation constant, $\lambda$ is the resonance wavelength, and $\text{Re}(n_{\text{eff}})$ is the real part of the mode effective index. At the resonance, minima is achieved in Eq. (1). This condition and the experimentally measured FSR makes it possible to estimate the mode effective index $\text{Re}(n_{\text{eff}})$ and the corresponding resonance order. The fitted curve results are shown as black circles in Figs. 5(b) and 6(a), from which we can see that there is good agreement between the simulation and experiment results. For the fitted transmission of the MKR [black circles in Fig. 5(b)], the estimated Re($n_{\text{eff}}$) = 1.47, $\gamma = 0.03$, and $\kappa_r = 0.29$, and its corresponding resonance order is 1445. For the fitted transmission of the MKR with SnS$_2$ [black circles in Fig. 6(a)], the estimated Re($n_{\text{eff}}$) = 1.42, $\gamma = 0.67$, and $\kappa_r = 0.21$, and its corresponding resonance order is 1393. The difference in the resonance order and mode effective index contributes to the resonance wavelength difference in the structure of the MKR with and without SnS$_2$. These simulated results indicate that absorption increases after the deposition of the SnS$_2$ nanosheets since the coupling loss changes from $\gamma = 0.03$ (MKR without SnS$_2$) up to $\gamma = 0.67$ (MKR with SnS$_2$). The simulated results also indicate that the absorption is increased after the deposition of SnS$_2$ since the coupling loss increases from $\gamma = 0.03$ (MKR without SnS$_2$) up to $\gamma = 0.67$ (MKR with SnS$_2$).

Quantitatively, the diameters of the MF and the MKR have hardly changed after coating with SnS$_2$, which shows the stability of the device [in Figs. 1(a) and 3]. Other resonance properties of the MKR with and without SnS$_2$ are summarized in Table 1. The resonance wavelength $\lambda_{\text{res}}$ of the maximum ER (18.0 dB) takes place at 1542.3 nm in the MKR without SnS$_2$. However, in the case of the MKR with SnS$_2$, the $\lambda_{\text{res}}$ of the maximum ER (26.6 dB) takes place at 1544.7 nm. The $\lambda_{\text{res}}$ of the maximum ER takes place at a larger wavelength, which might be due to the difference in the resonance order and its mode effective index, which is suggested by the fitted parameters of the resonance notch in Figs. 5(b) and 6(a) [6,34]. The resonance ER is higher in the MKR with SnS$_2$ than in the case of the MKR without SnS$_2$. This indicates that in the case of no SnS$_2$, the resonator might be at a state of undercoupling. However, in the case of the MKR with SnS$_2$, the resonator might be at a state of critical coupling, which will be further verified by the decreased ER phenomenon induced via the violet pump light demonstrated in the following. The change of the resonance condition in the MKR with SnS$_2$ might be due to the increased loss factor (estimated $\gamma = 0.67$ by coupled mode theory) brought by the SnS$_2$ nanosheets via light scattering and the absorption effect [34]. Thanks to the increase of ER, the MKR with the SnS$_2$ structure yields a larger resonance $Q$ ($\sim 59,415$) than that of the MKR without SnS$_2$ ($Q = 40,586$). In terms of the resonance FSR, it is 1.07 for the MKR without SnS$_2$, while for the MKR with SnS$_2$ it is 1.11. The small variation in FSR might be due to the small changes in the resonance conditions, such as coupling loss and coupling coefficients after coating with SnS$_2$ [6].

The violet pump light for the resonance amplitude tuning experiment is first performed on the MKR without the SnS$_2$ structure with the violet pump power varying at 0, 5.1, 10, 15.3, and 20.2 mW. The output transmission spectra are recorded, and the result at around 1542.3 nm is shown in Fig. 6(b). From Fig. 6(b), we can see that the transmitted light amplitude variation $\Delta T$ at the resonance dip wavelength $\lambda_{\text{res}}$ is smaller than 0.1 dB, and hardly no shift can be found in the $\lambda_{\text{res}}$. These results indicate that an MKR made of only silica-based MF cannot enable light amplitude tuning of the MKR resonance.

By employing the same experimental setup (Fig. 4) and the same violet light power variation, the violet pump light for resonance light amplitude tuning is then performed on the MKR with the SnS$_2$ structure. The measured output spectra around the wavelengths of 1544.7 nm and 1569.6 nm are shown in Figs. 7(a) and 7(b), respectively. With the increase of violet light power, the transmitted optical power increases.

### Table 1. Resonance Properties of Structures in the MKR with and without SnS$_2$

<table>
<thead>
<tr>
<th>Structure</th>
<th>$\lambda_{\text{res}}$ (nm)</th>
<th>$\text{ER}_{\text{max}}$ (dB)</th>
<th>$Q$</th>
<th>FSR (nm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>MKR without SnS$_2$</td>
<td>1542.3</td>
<td>18.0</td>
<td>40586</td>
<td>1.07</td>
</tr>
<tr>
<td>MKR with SnS$_2$</td>
<td>1544.7</td>
<td>26.6</td>
<td>59415</td>
<td>1.11</td>
</tr>
</tbody>
</table>

**Fig. 6.** (a) Measured normalized transmission spectra of the MKR with SnS$_2$ (green curve) and the corresponding fitted resonance curve (black circles). The fitted curve is obtained by setting $\gamma = 0.673$, $\kappa_r = 0.217$, and Re($n_{\text{eff}}$) = 1.42. (b) Transmission of the MKR structure at different external violet pump light powers. The red, black, brown, cyan, and blue curves correspond to the transmission with external violet pump power of 0, 5.1, 10, 15.3, and 20.2 mW, respectively.
corresponding to a decrease in the resonance ER. The largest $\Delta T$ of $\sim 4.5$ dB is obtained at a wavelength of 1544.7 nm, shown as the red ellipse in Fig. 7(a). The decreasing ER might indicate that the resonance condition changes from critical coupling to the undercoupling state under violet light excitation. The physical mechanism favoring this change of the resonance condition might probably be explained as follows: the strong absorption property of SnS$_2$ at 405 nm violet light will lead to the excitation of electron–hole pairs in SnS$_2$ nanosheets. These photon-generated carriers will then lead to both the real and imaginary parts of the index variation in SnS$_2$ nanosheets. As to the real part of the index variation in SnS$_2$ nanosheets, it relates and manifests as a wavelength shift in the resonance wavelength. However, no significant resonance wavelength shift can be found in Fig. 7. This might be due to the fact that the real part of the index variation is not big enough to induce a detectable mode effective index variation of the resonance. The detection of the resonance wavelength shift depends not only on the surrounded material index variations, but it also greatly relates to how sensitive the mode effective index is. On the other hand, the changes of the imaginary part of the SnS$_2$ nanosheet index might lead to variations in resonance conditions. Consequently, the transmitted power is varied accordingly. With the increase of the pump light power excitation, the concentration of the photon-excited carriers increases. This will then lead to an increase in the coupling loss factor for the MKR. For the resonance modes that are at the critical coupling state under no pump light excitation, an increase of the coupling loss factor will deviate the resonance state out of critical coupling [34]. Consequently, a decrease of the resonance ER or an increase of the transmitted power can be found at the vicinity of the resonance wavelength.

Within a signal light wavelength ranging from 1520 nm to 1620 nm, multiple resonances can be exploited for the amplitude tuning through the violet light absorption property by SnS$_2$ nanosheets. In order to investigate how the $\Delta T$ changes with respect to different resonance properties, we outline four different resonances (highlighted with red ellipses in Fig. 7) for detailed analysis. Correspondingly, the linear fit of $\Delta T$ versus violet light power for these resonances is shown in Fig. 8. Table 2 summarizes the resonance properties and the obtained $\Delta T$ variation rate associated with these resonances.

The largest $\Delta T$ variation rate with respect to violet light power is 0.22 dB/mW with a correlation coefficient of 98.4%, which corresponds to the black curve that has the steepest slope in Fig. 8. It is obtained at $\sim 1544.7$ nm, which corresponds to a resonance with the highest $Q$ (59,415) and the largest ER (26.6 dB), as is shown in bold font in Table 2. The second largest $\Delta T$ variation rate with respect to violet light power is 0.177 dB/mW with a correlation coefficient of 99.5% (pink curve in Fig. 8), which corresponds to a resonance with a lower $Q$ (20,652) and a smaller ER (19.2 dB) at $\lambda_{\text{res}}$ of 1569.6 nm, as is shown in italic font in Table 2. The other $\Delta T$ variation rates with respect to violet light power corresponding to resonances that have a poor $Q$ and a smaller resonance ER at 1533 nm with a correlation coefficient of 93.8% and at 1564 nm with a correlation coefficient of 98.4% are both 0.053 dB/mW. The above analysis for the four different highlighted resonance properties’ variation (in Figs. 7 and 8) under the same external violet light excitation shows a clear fact that resonances with higher $Q$ and larger ER can lead to higher sensitivities for the resonance properties’ variation rate. This can be understood by the fact that a resonance with a high $Q$ and ER will have a large amount of light energy stored inside the structure, which will then enhance the light–matter interaction. The enhanced light–matter interaction will then lead to a

**Table 2. Properties and the Obtained $\Delta T$ Variation Rate Associated with the Four Highlighted Resonances in Fig. 7**

<table>
<thead>
<tr>
<th>$\lambda_{\text{res}}$ (nm)</th>
<th>$Q$</th>
<th>ER</th>
<th>$\Delta T$ at 20.2 mW (dB)</th>
<th>$\Delta T$ Variation Rate (dB/mW)</th>
</tr>
</thead>
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<tr>
<td>1533</td>
<td>1915</td>
<td>3.7</td>
<td>1.0</td>
<td>0.053</td>
</tr>
<tr>
<td>1544.7</td>
<td>59415</td>
<td>26.6</td>
<td>4.5</td>
<td>0.22</td>
</tr>
<tr>
<td>1564</td>
<td>2016</td>
<td>4.2</td>
<td>1.1</td>
<td>0.053</td>
</tr>
<tr>
<td>1569.6</td>
<td>20652</td>
<td>19.2</td>
<td>3.7</td>
<td>0.177</td>
</tr>
</tbody>
</table>

**Fig. 7.** Transmission spectrum of the MKR with SnS$_2$ under different violet pump power excitation within a wavelength range of (a) 1532 nm to 1545 nm, while the two modes highlighted with red ellipses are around 1533 nm and 1544.7 nm, and (b) 1563 nm to 1570 nm, while the two modes highlighted with red ellipses are around 1564 nm and 1569.6 nm.

**Fig. 8.** Linear fit of $\Delta T$ versus violet light power for four different resonances at $\lambda_{\text{res}}$ = 1533 nm (red curve with a correlation coefficient of 93.8%), $\lambda_{\text{res}}$ = 1544.7 nm (black curve with a correlation coefficient of 98.4%), $\lambda_{\text{res}}$ = 1564 nm (blue curve with a correlation coefficient of 98.4%), and $\lambda_{\text{res}}$ = 1569.6 nm (pink curve with a correlation coefficient of 99.5%).
Therefore, this response time is a lower limit of the MKR with the system including the photodetector and the oscilloscope. The response time of the above measurements has already taken into account the response time of the measure-
ments is ∼ several tens of periods for different violet light powers. They all show good repeatability. The averaged rise time of the sample is ∼3.5 ms, and the averaged fall time is ∼3.7 ms among different measurements. The response time of the above measurements has already taken into account the response time of the system including the photodetector and the oscilloscope. Therefore, this response time is a lower limit of the MKR with the SnS2 structure. Further improvement such as developing a more homogeneous material deposition and better control over the deposition thickness might lead to a decrease of the response time.

Regarding sensitivity enhancement of the ∆T variation with respect to violet pump light excitation in the MKR coated with SnS2, the MKR without SnS2 has a ∆T of less than 0.1 dB under 20.2 mW violet pump light excitation, whereas the MKR with SnS2 yields a ∆T of 4.5 dB. As a consequence, the MKR with SnS2 has over a 45-fold enhancement in the ∆T variation under violet pump light excitation. Table 3 shows the performances of different types of light–control–light structures. In terms of sensitivity, the MKR with SnS2 demonstrated in this paper (bold font in Table 3) outperforms other configurations. The response time of the MKR with SnS2 yields a better result than those structures such as liquid crystals [35] and MoSe2 [36]. Consequently, the SnS2-coated MKRs demonstrated here, if they are further improved by optimized design for higher sensitivity and smaller response time, might be used as optical switches, multichannel amplitude modulators, and handheld fiber sensors. As an optical switch, the structure might need to be further optimized for internally pump light excitation in order to facilitate the control of switching on and off the resonance, and the response time should be reduced by improving the thickness homogeneity and the quality of SnS2 deposition. As multichannel amplitude modulators, an alternative geometry for adding the probe lights such as multiple monochromatic continuous waves at the corresponding MKR resonance wavelength might be a good choice. For the handheld fiber sensors, a more rigid assembling of the structure into a device should be developed.

4. CONCLUSION

In conclusion, we have demonstrated that by coating an MKR with 2D material SnS2 nanosheets, light–control–light functionality can be realized. Thanks to the multiresonance nature of the MKR, the sensitivity variations of resonances with different properties under the same external stimuli are demonstrated. It shows that a resonance of a higher Q and larger ER can lead to a higher sensitivity. The highest ∆T variation rate with respect to violet pump light power obtained in the MKR with the SnS2 structure is 0.22 dB/mW, corresponding to an MKR with a loop diameter of 480.6 μm that is made of a 7.0 μm diameter MF. It is obtained at λres around 1544.7 nm with a Q of 59,415 and an ER of 25.6 dB. In terms of response time, the structure can run as fast as ∼3.2 ms. The demonstrated light–control–light all-optical structure has the advantages of short response time, low cost, easy fabrication, and compatibility with fiber optics. Therefore, it might be a good candidate for developing fiber-compatible devices with other functionalities.

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